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# Multi-wavelength plasmonic optoelectronic memristor for reconfigurable logic operations and mixed-color pattern recognition

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**Abstract** Optoelectronic memristors have emerged as promising hardware platforms for highly efficient image perception owing to their low crosstalk, fast response, and low power consumption. However, the development of optoelectronic memristors with multi-wavelength recognition capabilities remains constrained by the limitations of their inherent bandgap. In this work, a multi-wavelength-modulated optoelectronic memristor that operates based on the localized surface plasmon resonance effect is developed using silver (Ag)-titanium dioxide (TiO<sub>2</sub>)/polymethyl methacrylate (PMMA) nanocomposites. The SET voltage can be gradually reduced by modulating the optical wavelength of visible irradiation. On this basis, reconfigurable IMP and False logic operations have been implemented in the memristive logic circuit. The neuromorphic visual preprocessing and recognition function for mixed-color patterns is demonstrated based on wavelength-dependent resistive switching behaviors. The proposed optoelectronic memristor enables high-density storage and in-memory computing.

**Keywords** optoelectronic memristor, localized surface plasmon resonance, multi-wavelength, resistive switching, color pattern recognition

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## 1 Introduction

With the rapid advancements in big data and artificial intelligence, devices based on the traditional von Neumann architecture are suffering from significant data delay and power consumption problems, mainly because of the separated storage and processing units [1–4]. Emerging neuromorphic devices have attracted great attention for emulating brain functions, such as learning and memory, exhibiting the potential to overcome the bottleneck observed in conventional devices [5–8]. Furthermore, memristors are considered the core foundation of neuromorphic computing hardware implementation because of their simple structure, fast response speed, low power consumption, and high data retention ability [9–12]. In particular, optoelectronic memristors integrate the benefits of a wide bandwidth and ultra-high speed, providing opportunities and advantages for photonic computing and neuromorphic vision [13–16].

To date, several photosensitive materials have been used to construct optoelectronic memristor devices. For example, Lu et al. [17] prepared a transverse optoelectronic memristor based on MAPbI<sub>3</sub> materials, achieving device conductance reduction under optical stimulation and thus simulating the long-term depression behavior of synapses. Hu et al. [18] reported a ZnO<sub>1-x</sub>/AlO<sub>y</sub> double-layer optoelectronic memristor with light recognition, memory, and processing capabilities, demonstrating sustained photoconductivity properties under the action of light signals. Zhao et al. [19] developed an optoelectronic memristor device based on N-doped carbon points, adjusted the SET voltage of the device using ultraviolet light, and performed optically controlled reconfigurable logic calculations. However, the inherent

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bandgap of photosensitive materials hinders their broad spectral response, which is not beneficial for mixed-color image recognition. Therefore, a novel optoelectronic memristor that can distinguish multi-wavelength light stimulation urgently needs to be developed.

The localized surface plasmon resonance (LSPR) effect, as an efficient interaction between light and matter, has attracted great interest in optoelectronic applications, including photocatalysis and optical detection [20, 21]. In LSPR, the optical response wavelength is dominated by the morphology and size of metal nanoparticles distributed on a material's surface [22, 23]. This tunability of the optical response can be exploited to overcome limitations associated with the inherent bandgap of the material. Therefore, leveraging the LSPR effect in an optoelectronic memristor may be an effective strategy for constructing a multi-wavelength optoelectronic memristor. In particular, constructing a metal/oxide interface with the appropriate potential barrier is necessary to achieve efficient separation of photogenerated electron-hole pairs [24, 25]. For an electrochemical metallization device, the oxidation of metal nanoparticles induced by the LSPR effect dominates the ion migration process under an electric field. The resulting multi-wavelength-modulated device is useful for constructing an artificial neural network (ANN) that can perform mixed-color pattern recognition. However, plasmonic optoelectronic memristors are still in their infancy, and few investigations have been reported.

In this study, we propose a multi-wavelength optoelectronic memristor based on silver (Ag)-titanium dioxide (TiO<sub>2</sub>)/polymethyl methacrylate (PMMA) nanocomposite films using the LSPR effect. The resistive switching behaviors can be modulated using multi-wavelength light irradiation, in which the SET voltage gradually decreases. The formation/annihilation mechanism of Ag conductive filaments (CFs) has been verified by conductive atomic force microscopy (C-AFM) and temperature dependence measurements. Additionally, several logic operations have been demonstrated in the memristive circuits, including IMP and False gates. Furthermore, a mixed-color image recognition function has been implemented in a simulated ANN based on the optoelectronic memristor. The mixed-color image tests reveal a high recognition rate of 96.4%. The proposed multi-wavelength optoelectronic memristor has significant potential for application in efficient neuromorphic visual systems.

## 2 Experimental

**Synthesis of Ag-TiO<sub>2</sub>/PMMA composites.** First, 0.5 g TiO<sub>2</sub> was immersed in 4 mol/L AgNO<sub>3</sub> solution and ultrasonicated for 2 h. Then the mixed dispersion was transferred to a petri dish and irradiated with UV light (310 nm). After a certain period of irradiation, the mixture was centrifuged, and pour out the supernatant. The remaining solids were centrifugally cleaned four times using deionized water, then transferred to a slide and placed in an oven at 60°C for 6 h. PMMA was dissolved in trichloromethane at a concentration of 5 mg/mL and ultrasonicated for 2 h. Then, different amounts of Ag-TiO<sub>2</sub> were added to the PMMA solution. The final Ag-TiO<sub>2</sub>/PMMA suspensions with different concentrations were obtained by stirring in a centrifugal tube for 6 h.

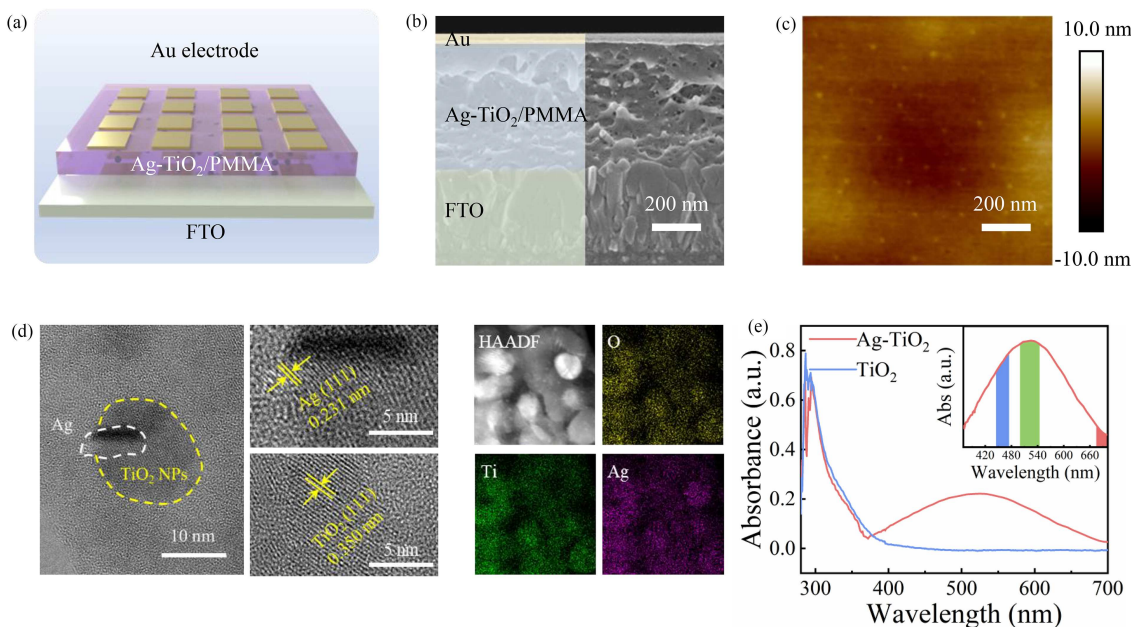
**Fabrication of the optoelectronic memristive device.** Fluorine-doped tin oxide (FTO) glass was used as the substrate and the bottom electrode material. The Ag-TiO<sub>2</sub>/PMMA suspensions were spun onto the FTO substrates at room temperature. Finally, the top Au electrodes were deposited by electron-beam evaporation.

**Electrical measurements.** The electrical measurements were performed using a source meter (2636A, Keithley). The optical modulation was performed using a xenon lamp (LA-410UV, Hayashi).

**Implementation of spike-timing-dependent-plasticity (STDP).** To implement STDP using Ag-TiO<sub>2</sub>/PMMA-based plasmonic optoelectronic memristors, a series of single pulses were consecutively applied to the top electrode and bottom electrode. The first pulse was a negative voltage; then, positive pulses with decreasing amplitudes were subsequently applied. The timing interval  $\Delta t > 0$  ( $\Delta t < 0$ ) was defined as the interval between the end of the pre-synaptic (post-synaptic) spikes and the beginning of the post-synaptic (pre-synaptic) spikes. The post-synaptic conductance was measured in the dark state ( $G_0$ ) and after 10 min visible light irradiation ( $G_1$ ) with the paired spike. This represents the long-term plasticity of the synapse. The change in synaptic weight  $\Delta W$  is defined as  $(G_1 - G_0)/G_0$ .

**Calculation of accuracy.** Structural similarity (SSIM) was employed to measure the difference between the learned image ( $x$ ) and the input image ( $y$ ). Here, high SSIM represents high learning accuracy. SSIM can be expressed as follows:

$$\text{SSIM}(x, y) = f[l(x, y), c(x, y), s(x, y)], \quad (1)$$



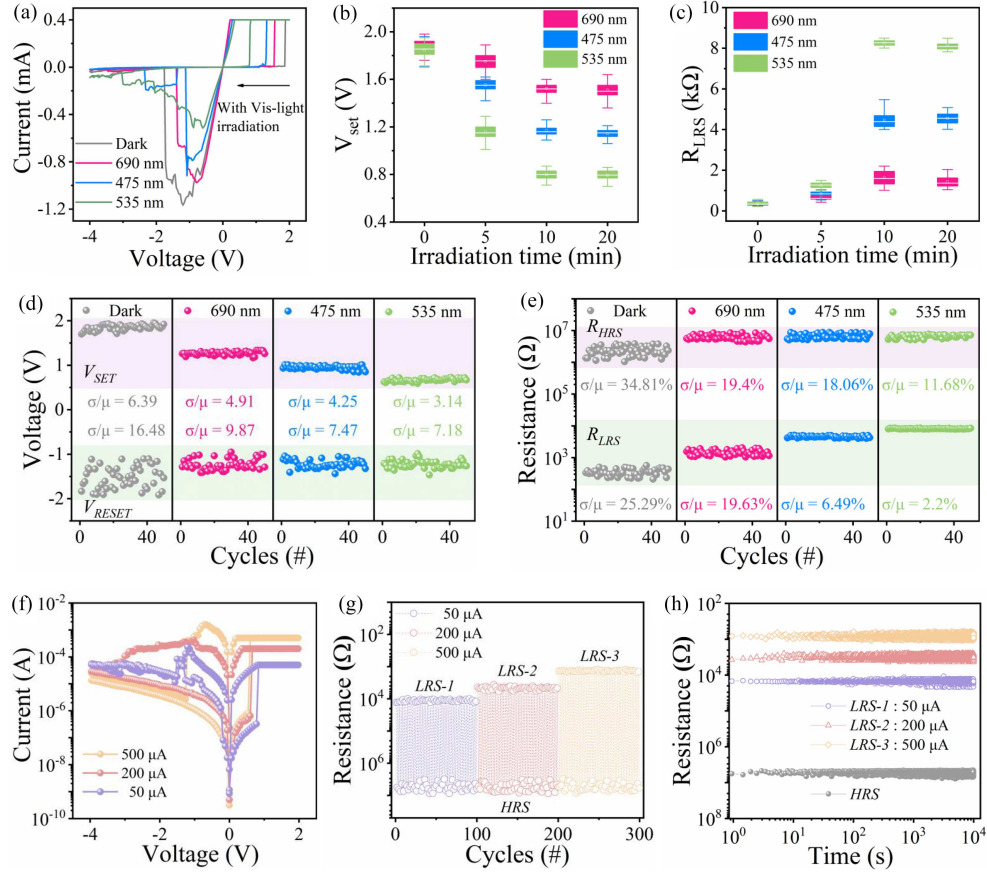
**Figure 1** (Color online) Structural characterization of plasmonic optoelectronic memristor based on Ag-TiO<sub>2</sub>/PMMA nanocomposite. (a) Illustration of the device, including Au top electrodes, Ag-TiO<sub>2</sub>/PMMA films, and FTO bottom electrodes; (b) cross-sectional SEM image of the Au/Ag-TiO<sub>2</sub>/PMMA/FTO memory device; (c) AFM image of the Ag-TiO<sub>2</sub>/PMMA layer; (d) HRTEM images of the Ag-TiO<sub>2</sub> nanocomposite structures and elemental mapping images of the Ag-TiO<sub>2</sub>/PMMA film; (e) absorption spectra of the Ag-TiO<sub>2</sub> film and a pure TiO<sub>2</sub> film. The inset shows the differential spectrum.

where  $l(x, y)$ ,  $c(x, y)$ , and  $s(x, y)$  are the brightness, contrast, and structure of the image, respectively. A detailed explanation of this metric can be found in [26].

### 3 Results and discussion

Figure 1(a) shows an illustration of the proposed plasmonic optoelectronic memristor device with an Au/Ag-TiO<sub>2</sub>/PMMA/FTO structure. The Ag-TiO<sub>2</sub> nanoclusters were prepared by photocatalytic reduction in AgNO<sub>3</sub> solution, and the functional layer was fabricated by spin-coating (see more details in Section 2). As shown in Figure 1(b), the cross-sectional scanning electron microscopy (SEM) image indicates that the thickness of the Ag-TiO<sub>2</sub>/PMMA nanocomposite layer is 650 nm. Corresponding top-view SEM images are shown in Figure S1(a). The top Au electrode with a diameter of 500 μm is deposited by high vacuum thermal evaporation. In addition, the surface morphology of the Ag-TiO<sub>2</sub>/PMMA film has been characterized by AFM, as shown in Figure 1(c), confirming the high film quality. The Fourier transform infrared spectrum of Ag-TiO<sub>2</sub>/PMMA is shown in Figure S1(b). The stretching vibrations of C=O and C-O bonds appear at 1716 and 1096 cm<sup>-1</sup>, respectively. The stretching vibrations of C-H bonds in methyl and methylene groups show adsorption peaks at 2923 and 3008 cm<sup>-1</sup> [27, 28]. The high-resolution transmission electron microscopy (HRTEM) images in Figure 1(d) further confirm that Ag and TiO<sub>2</sub> nanoparticles form nanocomposite structures. The lattice fringes at 0.231 and 0.350 nm correspond to the (111) plane of Ag and the (101) plane of anatase TiO<sub>2</sub>, respectively [29, 30]. Furthermore, the right panel in Figure 1(d) shows the corresponding energy-dispersive X-ray mapping spectra, displaying the compositional distribution. To further investigate the structure of the Ag-TiO<sub>2</sub> nanoparticles, X-ray diffraction was performed, as shown in Figure S1(c). Diffraction peaks have been obtained for both TiO<sub>2</sub> (anatase phase) and Ag, further confirming the successful introduction of Ag in the composite film [31, 32]. Figure 1(e) shows the absorption spectra of TiO<sub>2</sub> and Ag-TiO<sub>2</sub> thin films. Compared with the TiO<sub>2</sub> films, a distinct absorption peak is observed in the visible region (centered at 535 nm) in Ag-TiO<sub>2</sub> films. According to the literature, this absorption peak can be attributed to the LSPR absorption properties of Ag nanoparticles [33–37].

The effects of visible light irradiation on the resistive switching (RS) behavior are investigated first. The devices with Ag-TiO<sub>2</sub> concentrations of 0.5 wt%, 1.0 wt%, and 1.5 wt% exhibit an abrupt transition between the high resistance state (HRS) and low resistance state (LRS) during the voltage sweep, as



**Figure 2** (Color online) Effects of visible light irradiation on resistive switching behaviors. (a) Typical RS behaviors after visible light irradiation of different light wavelengths. The values for (b)  $V_{SET}$  and (c) LRS resistance of different devices under visible light irradiation of different wavelengths. The data are collected from 50 consecutive RS cycles for each point after exposure to the given irradiation for 10 min. (d) The statistical results of the  $V_{SET}/V_{RESET}$  and (e)  $R_{HRS}/R_{LRS}$  values under continuous RS measurements after visible light irradiation of different light wavelengths. (f) Typical  $I$ - $V$  characteristics of the device under CCs of 50, 200, and 500  $\mu$ A. (g) Resistance values and (h) retention characteristics of multilevel resistance states with different CC values.

shown in Figure S2. To optimize the operation, the Ag-TiO<sub>2</sub> concentration of 1.5 wt% is selected for subsequent investigation. The  $I$ - $V$  characteristics of Au/Ag-TiO<sub>2</sub>/PMMA/FTO memory cells subjected to different wavelengths (red, 690 nm; green, 535 nm; and blue, 475 nm; with a power of 36 mW cm<sup>-2</sup>) are measured, as shown in Figure 2(a). For the initial state without visible light exposure, a forming process is required to activate the device by applying a positive voltage (2.95 V) to the top electrode, and the device then exhibits typical bipolar resistive behavior (Figure S2). In contrast, the device abruptly switches on at the SET voltage of 0.8 V after employing visible light irradiation, indicating that the forming process is not required. Figures 2(b) and (c) show the  $V_{SET}$  and  $R_{LRS}$  values of the device during continuous cycle operation after visible light irradiation of different durations (0, 5, 10, and 20 min) and wavelengths (red, 690 nm; green, 535 nm; and blue, 475 nm; with a power of 36 mW cm<sup>-2</sup>). Herein,  $V_{SET}$  decreases from 1.87 to 1.52, 1.13, and 0.79 V under visible light irradiation of 690, 475, and 535 nm, and  $R_{LRS}$  increases from  $3 \times 10^2$  to  $1.5 \times 10^2$ ,  $4.5 \times 10^3$ , and  $8.1 \times 10^3 \Omega$ , respectively. Furthermore, when the irradiation is applied for 10 min, the  $R_{LRS}$  ( $V_{SET}$ ) value does not continue to increase (decrease), which may be due to saturation of the photoinduced oxidation in Ag nanoparticles. Therefore, we chose 10 min as the optimum pretreatment time. Figures 2(d) and (e) show the  $V_{SET}/V_{RESET}$  and  $R_{HRS}/R_{LRS}$  values collected during 50 consecutive switching cycles after visible light irradiation.  $V_{SET}/V_{RESET}$  and  $R_{HRS}/R_{LRS}$  exhibit negligible fluctuations, which means the stability of Au/Ag-TiO<sub>2</sub>/PMMA/FTO memory devices can be improved after visible light irradiation. Furthermore, a more distinct increase/decrease in  $R_{LRS}$  and  $V_{SET}$  is obtained after visible light irradiation. The influence of the irradiation wavelength is consistent with the light absorption characteristics of the Ag-TiO<sub>2</sub>/PMMA nanocomposite films. It is worth noting that the data overlap between these SET voltage values is not very significant, as shown in Figure S3. The

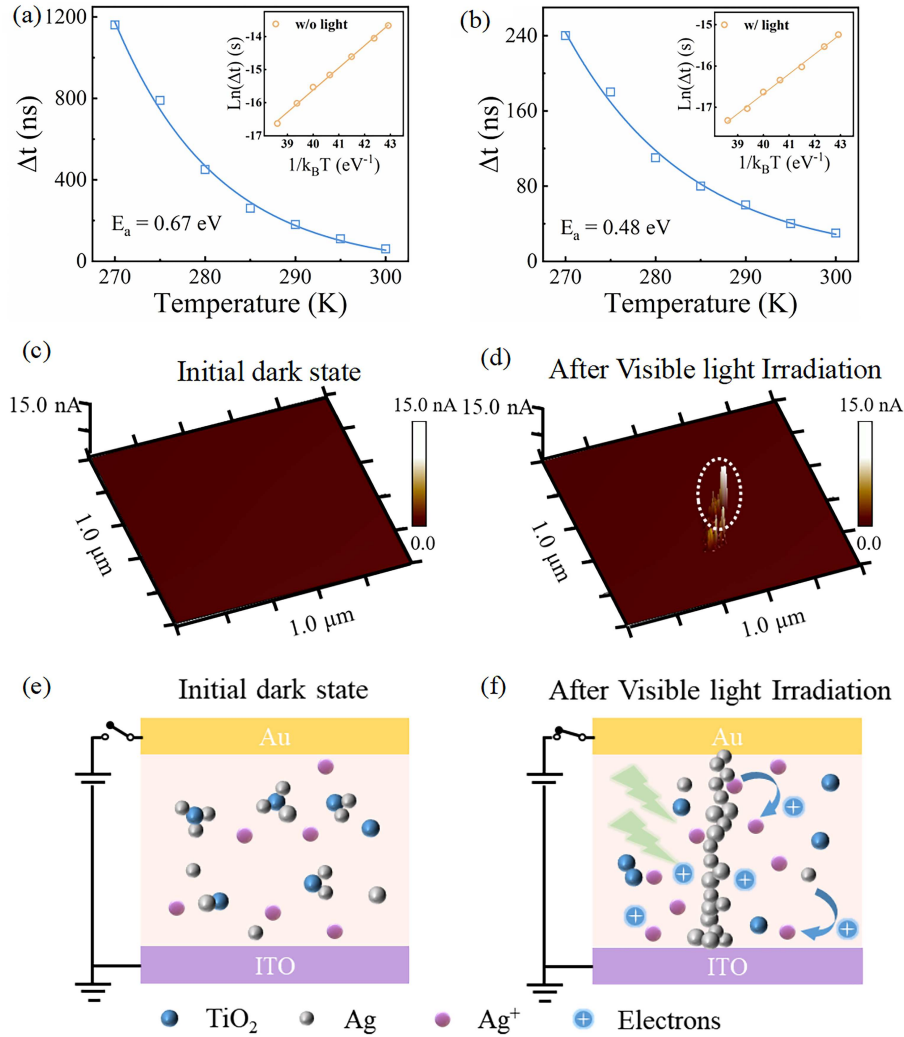


stable characteristics of device to device provide a guarantee for mixed-color image recognition. Moreover, multilevel storage behavior has been demonstrated by applying compliance currents (CCs) of 50, 200, and 500  $\mu\text{A}$ . Figure 2(g) shows the differentiated resistance distributions obtained for 50 consecutive on-off cycles. As shown in Figure 2(h), all four resistance states exhibit negligible degradation within  $10^4$  s, indicating high retention ability. In addition, the mechanical flexibility of the memristor has been evaluated, and the electrical response of the memristor under different bending strains is shown in Figure S4. No significant degradation was observed in the resistivity under different bending strains, indicating that the Au/Ag-TiO<sub>2</sub>/PMMA/FTO memory unit has effective bending resistance. Overall, our device presents broad application prospects in flexible neuromorphic vision systems.

We also consider the underlying memristive mechanism for multi-wavelength-modulated RS behaviors in our Ag-TiO<sub>2</sub>/PMMA-based device. In Ag-TiO<sub>2</sub>/PMMA nanocomposite films, the LSPR effect under visible light irradiation is accompanied by the generation of Ag<sup>+</sup>, and Ag<sup>+</sup> migration occurs in an electric field. According to the ion migration model, the Ag<sup>+</sup> migration process is activated by heat, and the switching time is proportional to  $\exp(-E_a/k_B T)$ , where  $E_a$  is the activation energy of ion migration,  $k_B$  is the Boltzmann constant, and  $T$  is temperature [38, 39]. Herein, Ag-TiO<sub>2</sub>/PMMA memory cells in the initial state and after 10 min of visible light irradiation were used for comparison. Temperature-dependent SET measurements are performed in the pulsed program mode (as shown in Figure S5(a)), with and without visible light exposure for 10 min, to determine  $E_a$  values. The SET curves in the initial state and after 10 min of irradiation are shown in Figure S5, and the corresponding switching times ( $\Delta t$ ) depending on temperature are shown in Figures 3(a) and (b), respectively. A linear relationship between the  $\ln(\Delta t)$  and  $1/k_B T$  curves is obtained, for which the  $E_a$  values are 0.67 and 0.48 eV in the initial dark state and after 10 min of visible light irradiation, respectively. Therefore, the LSPR effect reduces the migration barrier of Ag<sup>+</sup>, which is responsible for the lower  $V_{\text{SET}}$  value [40, 41]. To better understand the physical mechanism of the light-modulated RS behavior, we perform C-AFM, as shown in Figures 3(c) and (d). In the initial dark state, the entire surface area exhibits a relatively low current value, representing the HRS. In a dark environment, a locally bright region is obtained after applying a SET voltage, indicating the formation of strong CFs (Figure S6). Under visible light irradiation at 535 nm, a thinner CF is formed, as shown in Figure 3(d). On this basis, a possible switching model is proposed, as shown in Figures 3(e) and (f). In the LRS, Ag CFs are formed through the oxidation, migration, and reduction of Ag under an electric field. Meanwhile, the LSPR effect effectively promotes the generation of Ag<sup>+</sup> and reduces the SET voltage when exposed to visible light, thus inhibiting the uncontrollable growth of CFs.

Next, memristive logic circuits have been designed to further evaluate the application of our device. As depicted in Figure 4(a), the IMP logic circuit consists of two RRAM devices, namely P and Q, which are connected to the ground using a  $10^4 \Omega$  resistor  $R_L$  ( $\text{LRS} \ll R_L \ll \text{HRS}$ ). Here, we define two beams of incident light with intensities of 0 and 36  $\text{mW cm}^{-2}$  as input light 0 and 1, respectively. The current values corresponding to HRS and LRS (monitored by electrical pulse 0.2 V/1 ms) are defined as logical output 0 and 1, respectively. Considering the switching voltage and the voltage distribution over the  $R_L$ , the electrical pulses applied to devices P and Q are set as 0.9 V/1 ms and 2 V/1 ms, respectively. For the IMP operation, devices P and Q are initialized to HRS. In the input process, the method of simultaneous input is adopted. After two beams of visible light are irradiated,  $V_P$  (0.9 V/1 ms) and  $V_Q$  (2 V/1 ms) are simultaneously applied to the memristor. Then the output results are read by electrical pulses of 0.2 V. If the devices P and Q are written 0 and 0, the device Q will change to LRS after the IMP operation, obtaining the logic 1. If the devices P and Q are written 1 and 0, the voltage drop on device Q cannot make the device switch from the HRS to LRS when device P is in the LRS. When the devices P and Q are written 1 and 1, the devices P and Q are almost simultaneously transformed into LRS, resulting in the output of 1. The logic output is consistent with the truth result of IMP, indicating that the IMP operation has been successfully implemented (Figure 4(b)). Figure 4(c) shows the False logic operation achieved with the assistance of 475 and 690 nm light ( $36 \text{ mW cm}^{-2}$ ). For False logic instructions, devices P and Q cannot switch to the LRS by applying  $V_P$  (0.9 V/1 ms) or  $V_Q$  (0.9 V/1 ms) under 475 and 690 nm light. Then, the output results are read by electrical pulses of 0.2 V. The measured logic results and true values are shown in Figure 4(d); the output states can be stably retained and stored, enabling non-volatile stateful reconfigurable logic.

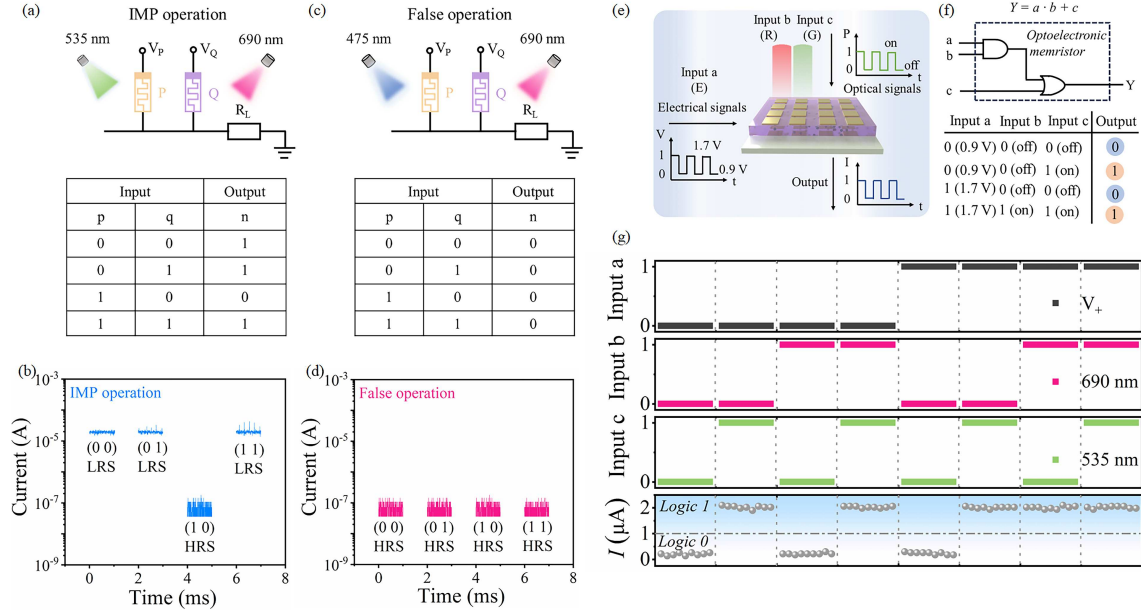
On the basis of the variation in the photoresistive characteristics of the memristor, time-resolved current measurements are carried out at two wavelengths and under positive bias, as shown in Figure 4(e). In this configuration, the electrical signal and the optical signals of 690 and 535 nm serve as input signals  $a$ ,



**Figure 3** (Color online) RS mechanism of Au/Ag-TiO<sub>2</sub>/PMMA/FTO memristors. (a) and (b) Temperature dependence of SET switching time for the memristor without and with 535 nm light irradiation. The Arrhenius plots used for determining the migration barrier energy are shown in the insets. Local conductivity of the Au/Ag-TiO<sub>2</sub>/PMMA/FTO memristor measured by C-AFM (10 mV bias) for (c) the initial state and (d) the LRS after SET operation using 535 nm light irradiation. Illustrations of the switching mechanism of the Au/Ag-TiO<sub>2</sub>/PMMA/FTO memory devices in (e) the initial state and (f) the LRS after SET operation with 535 nm light irradiation.

*b*, and *c*, respectively. For electrical input, the input conditions of 0.9 V/1 ms and 1.7 V/1 ms represent logic 0 and 1, respectively. As for the optical input (690 and 535 nm), the off and on states are set as 0 and 1, respectively. The current values corresponding to the HRS and LRS of the optoelectronic memristor are defined as 0 and 1 (denoted as *Y*), respectively. Output currents above and below a threshold current of 1 μA are defined as 1 and 0, respectively. As shown in the truth table (Figure 4(f)), when the input signals of *a*, *b*, and *c* are 00001111, 001110011, and 01010101, the output signal of *Y* is 01010111. Therefore, the logic operation can be achieved in a single device (Figure 4(g)). In previous studies, at least two transistors were required for the logic operations [42–45]. Consistent with our previous results, the multi-wavelength plasmonic optoelectronic memristor exhibits great potential in reconfigurable logic circuits.

To demonstrate visual multifunctional integration, we design a neuromorphic vision system using an optoelectronic memristor to implement image preprocessing and pattern recognition (Figure 5(a)). In this system, the optoelectronic memristor array serves as the preprocessing module to extract and store wavelength information. Meanwhile, the memristive neural network performs the image recognition function, to recognize the red, green, and blue (RGB) information. Figure 5(b) illustrates the preprocessing results for the RGB mixed-color images (represented using numbers 1–6). The multi-wavelength information can be distinguished using different SET voltages. As plotted in the left panel, only the memristive device

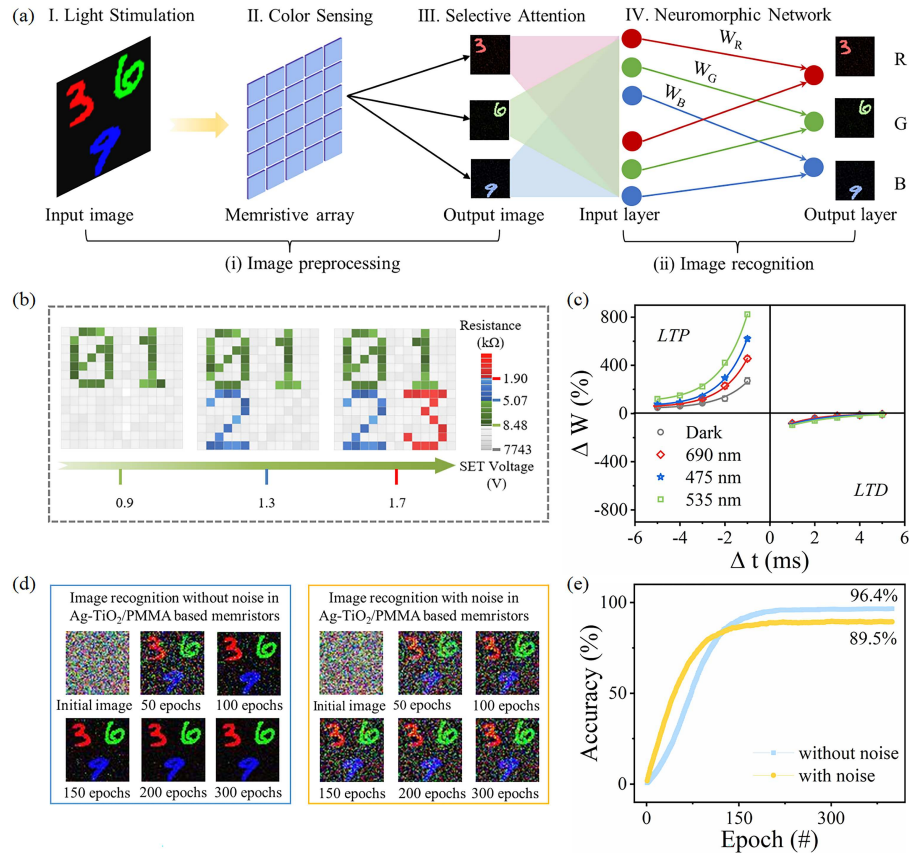


**Figure 4** (Color online) Optoelectronic logic operating with multiple optical input sources. (a) Circuit and truth table for IMP operation. (b) Readout current of RRAM devices P and Q after the IMP operation. (c) Circuit and truth table for False operation. (d) Readout current of devices P and Q after False operation. (e) and (f) Basic operation principle and schematic of the logic circuit for logical operation based on Ag-TiO<sub>2</sub>/PMMA-based plasmonic optoelectronic memristor devices. The nonlinear output current is programmed by 690 and 535 nm light signals and electrical pulses. (g) Output current responses of the optoelectronic three-input logic application.

irradiated with green light can be switched to the LRS under the SET voltage of 0.9 V. In contrast, the SET voltage of 1.7 V is required to realize RS behavior under red light irradiation. Then, the feature information is sent to the ANN for subsequent image recognition. Note that the STDP rule is adopted as a common learning rule to implement color image recognition [46, 47]. For the STDP learning rule, the synaptic weight changes ( $\Delta W$ ) are dominated by the interval between pre-synaptic and post-synaptic spikes. Herein, the synaptic weight is defined as  $\Delta W = (G_1 - G_0)/G_0$ , where  $G_0$  and  $G_1$  represent the initial and stable conductance after electrical pulses, respectively. Detailed information regarding the STDP measurement is presented in Section 2. As shown in Figure 5(c), visible light irradiation enhances the synaptic weight change to a different extent, in which the greater  $\Delta W$  is obtained with green light irradiation. Accordingly, mixed-color image recognition is demonstrated in our memristive neural network. The input image is composed of  $56 \times 56$  pixels. Figure 5(d) shows the evolution of images with different training numbers during the original transformation of the image and the learning process after using Gaussian noise. The dependence of the recognition accuracy with and without Gaussian noise on the number of training epochs is plotted in Figure 5(e). After 400 training epochs, the learning accuracy of the neural network based on the Ag-TiO<sub>2</sub>/PMMA photoelectric memristor device reaches 96.4%. Meanwhile, the recognition accuracy can still exceed 89.5% under Gaussian noise with a mean of 0 and a variance of 0.1. In addition, different noise levels are used to demonstrate its anti-interference ability, as shown in Figure S8. The results show that the feature information is effectively extracted during the preprocessing of the optoelectronic memristor array, resulting in high-precision color pattern recognition.

## 4 Conclusion

In this study, we have developed a multi-wavelength identifiable memristor device based on Ag-TiO<sub>2</sub>/PMMA nanocomposites. In particular, multilevel storage has been realized using visible light irradiation of different wavelengths. The multi-wavelength modulation characteristics can be attributed to the photoinduced oxidation of Ag nanoparticles. The photoinduced oxidation process promotes Ag<sup>+</sup> migration under an electric field and reduces the SET voltage. Furthermore, IMP and False logic operations have been demonstrated by combining electrical and optical signals in our memristive circuits. Finally, neuro-morphic preprocessing and recognition of mixed-color images have been realized in simulated ANNs. The



**Figure 5** (Color online) Mixed-color pattern recognition in the Ag-TiO<sub>2</sub>/PMMA optoelectronic memristor-based neuromorphic visual system. (a) Illustration of the neuromorphic vision system, which contains the image preprocessing and recognition parts based on visual attention and STDP synaptic weight updating; (b) color recognition preprocessed by the memristor array; (c) synaptic weight variation ( $\Delta W$ ) versus relative spike timing ( $\Delta t$ ), showing typical STDP characteristics, similar to biological synapses; (d) evolution of the images during the learning processes without noise and with noise in Ag-TiO<sub>2</sub>/PMMA-based memristors; (e) recognition accuracy as a function of the learning epochs used for the memristive ANN.

neuromorphic preprocessing function reduces redundant data and provides high recognition accuracy. The proposed plasmon optoelectronic devices show great prospects in the field of logic calculation and neuromorphic vision.

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**Supporting information** Figures S1–S8. The supporting information is available online at [info.scichina.com](http://info.scichina.com) and [link.springer.com](http://link.springer.com). The supporting materials are published as submitted, without typesetting or editing. The responsibility for scientific accuracy and content remains entirely with the authors.

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## Profile of Yichun LIU



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